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begradable polymer matrices and porous scaffolds provide powerful mechanisms for newsive, sustained release of drugs relevant to the treatment of a broad range of diseases **t** he needs of the patient via programmable engineering platforms that include power **s**ources, delivery mechanisms, communication hardware, and associated electronics, be most typically in forms that require surgical extraction after a period of use. Here we <br/>report a light-controlled, self-powered technology that bypasses key disadvantages <br/> of these systems, in an overall design that is bioresorbable. Programmability relies **u** on the use of an external light source to illuminate an implanted, wavelength-sen-<br/>sitive phototransistor to trigger a short circuit in an electrochemical cell structure that includes a metal gate valve as its anode. Consequent electrochemical corrosion network the gate, thereby opening an underlying reservoir to release a dose of drugs <br/>by passive diffusion into surrounding tissue. A wavelength-division multiplexing net te de la company de la compa tion of a collection of reservoirs built into an integrated device. bioresorbable electrode materials define the key considerations and guide optimized **c**hoices in designs. In vivo demonstrations of programmed release of lidocaine adja- cent the sciatic nerves in rat models illustrate the functionality in the context of pain nanagement, an essential aspect of patient care that could benefit from the results presented here.

bioresorbable | self-powered | battery | drug delivery | light-controlled

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 Passive release of drugs loaded into polymer matrices (1–4) or porous scaffolds (5–8) addresses essential needs in patient into polymer ratices (1–4) or porous scaffolds (5–8) addresses essential needs in patient into polymer matrices (1–4) or porous scaffolds (5–8) addresses essential needs in patient into polymer matrices (1–4) or porous scaffolds (5–8) addresses essential needs in patients into polymer matrices (1–4) or porous scaffolds (1–5) addresses essential needs in polymer and text is repeated in the patient of the patient into processes essential needs in patients in the patient into processes essential needs in patients in the processes essential needs in patients in the processes essential needs in the needs in the processes essential needs in the needs in the processes essential needs in the n

### Significance

Polymer matrices and porous scaffolds enable passive, gradual release of drugs, with relevance to important clinical applications. Actively controlled systems represent an appealing alternative due to capabilities in user-defined programmability over the timing of discrete release events. Nevertheless, requirements for supporting power supplies and wireless electronics, together with the need to surgically extract devices after a period of use, represent disadvantages that limit broad adoption. This work addresses this opportunity through a self-powered, light-controlled, bioresorbable platform that bypasses the need for separate power supplies and enables natural dissolution and clearance from the body after treatment. This miniaturized and wirelessly programmed drug delivery device represents a significant advance over previously reported technologies, with potential for pharmacological treatment of a broad range of diseases.

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Fig. 1. Self-powered, light-controlled, bioresorbable devices for programmed drug release. Schematic illustrations of (A) the process for implanting a device for a site of interest, (B) the method for light-controlled, bioresorbable devices for programmed drug release. Schematic illustrations of (A) the process for implanting a device at a site of interest, (B) the method for light-controlled, bioresorbable devices for programmed be release of drugs at a site of a site of a site of interest, (B) the method for light-controlled, bioresorbable devices for programmed be released of the site of a site of a site of a site of a site of site of

#### Results

Device Designs. Fig. 1D and SI Appendix, Fig. S1 summarize the mechanisms for self-powered operation and the scheme for remote, programmable control. In this example, a set of three electrochemically active, bioresorbable metal for a more electropower of a control of a more electropower of solated anodes. A foil of a more electropower of solated encodes as a single cathode. A molded structure of a bioresorbable polyanhydride (29) defines discrete reservoirs, each filled with a drug and sealed by an anode. Separate sepa

collector terminals of the phototransistors. The interconnect structure allows drug reservoirs to be implanted to the sites of interest, including at depth below the surface of the skin, with phototransistors located at shallow locations below the skin, to facilitate illumination by an external light source. The implantation in the body forms a battery with a collection of electrochemical cells, one for each reservoir, where the biofluid serves as a common electrolyte. Exposing a phototransistor to light decreases its resistance and thus short circuits the corresponding cell. The result dramatically accelerates the corrosion of the anode, to open the underlying reservoir and allow release of drug into the surrounding tissues by passive diffusion. The device is self-powered because the drug reservoir structure itself acts as a battery, to provide power needed to open the gates. The phototransistor serves only as a passive switch, in the sense that exposure of this component of the system does not provide power for operating the device. Optical filters placed atop the phototransistors provide the basis for wavelength-selective activation. This scheme allows for independent control of each reservoir in the array, in a form of wavelength-division multiplexing. In the example Fig. 1E shows an exploded view schematic illustration of the design. For cases reported here, the anode gates use Mg, Mg coated with spin-on-glass (Mg-SOG), or Zn. The cathodes use Fe, Mo, W, or MoO<sub>3</sub>. Fig. 1 F and G and SI Appendix, Fig. S3 present photographs of a device with 5 mm  $\times$  5 mm Mg-SOG anodes (50  $\mu$ m thick, 3 mm × 3 mm gate opening) and a 5 mm × 17 mm Fe cathode (25  $\mu$ m thick). Each reservoir has dimensions of 5 mm  $\times$  13 mm and a depth of 1 mm, which corresponds to a drug volume of  $\sim 65 \mu$ L. The degradation rates of device materials can be affected by the pH, the presence of enzymes, and processes related to immune responses, as well as the temperature (SI Appendix, Table S2). The rates in all cases increase with temperature. The corrosion rates of Mg and Fe increase with decreasing pH (31, 32). By contrast, the <br/> degradation rates of Mo, SOG, and PA increase with increasing pH because OH<sup>-</sup> initiates or facilitates the degradation reaction (29, 33, 34). Enzymes have minor or no impact on the degradation of Mg (35) and PA (36, 37). Fibrotic capsules that result from the immune response suppress the rate of corrosion of Mg (38) and may also decrease the degradation rates for other device components as a barrier to diffusive transport of biofluids and reaction products. Overall, the precise rate of degradation does not critidetermine the utility of the device, as long as it disappears eventually, in a timeframe between a few months and a few years.

Fig. 1H and SI Appendix, Fig. S4 display results of accelerated aging tests that involve submerging a system without the interconnects and phototransistors/filters in phosphate-buffered saline (PBS) (pH 7.4) at 95 °C. Complete dissolution of 50-µm-thick Mg-SOG anodes, 250-µm-thick PA (1:1:2.5) encapsulation, and a 5-µm-thick Mo cathode occurs within 4, 17, and 33 d, respectively. The biodegradation mechanisms and rates of device materials, as well as their biological fates of the products of degradation, are summarized in SI Appendix, Fig. S5 and Tables S3 and S4. These data indicate that the device can be eliminated from the body (in vivo) after ~4 to 5 mo of implantation. The products of degradation of Mg, SOG, Mo, and PA are Mg(OH)<sub>2</sub>, silicates, H<sub>2</sub>MoO<sub>4</sub>, and carboxylic acids, respectively. Each of these products is water soluble and is harmlessly excreted in the urine (26, 39–42). Fe degrades in a spatially nonuniform manner, with products (iron hydroxides) that have low solubility in biofluids but without known adverse biological effects (26). In this sense, Mo might be preferred to Fe for certain applications. H<sub>2</sub>, as an additional degradation product of Mg, dissolves into the adjacent tissues, diffuses through these tissues and blood vessels, and can pass through the skin (43). Certain of these materials including Mg and PA are widely used in biodegradable, nonelectronic implants at significantly larger amounts, as further evidence of an understanding of their biocompatibility.

**Choice of Gate and Battery Materials.** Fig. 2*A* lists materials options for the bioresorbable electrodes, along with their main reactions with surrounding biofluids and standard electrochemical potentials. As mentioned above, the anodes include Mg or Zn, paired with cathodes of Fe, Mo, W, or MoO<sub>3</sub>. The corresponding reactions that govern bioresorption in an aqueous environment



Fig. 2. Characteristics of bioresorbable gates and battery materials. (A) Electrochemical reactions and standard potentials of bioresorbable gates and battery materials. (A) Electrochemical reactions and standard potentials of bioresorbable gates and battery materials. (A) Electrochemical reactions and standard potentials of bioresorbable gates and battery materials. (A) Electrochemical reactions and standard potentials of bioresorbable gates and battery materials. (A) Electrochemical reactions and standard potentials of bioresorbable gates and battery materials. (A) Electrochemical reactions and standard potentials of bioresorbable gates and battery materials. (A) Electrochemical reactions and standard potentials of bioresorbable gates and battery materials (A) Electrochemicals and standard potentials of bioresorbable gates and battery materials (A) Electrochemicals and standard potentials of the standard potential of the standard potenti

are: Mg + 2H<sub>2</sub>O → Mg(OH)<sub>2</sub> + H<sub>2</sub>; Zn + 2H<sub>2</sub>O → Zn(OH)<sub>2</sub> + H<sub>2</sub>;  $4Fe + 6H_2O + 3O_2 \rightarrow 4Fe(OH)_3$ ;  $2Mo + 2H_2O + 3O_2$  $\rightarrow 2H_2MoO_4$ ;  $2W + 2H_2O + 3O_2 \rightarrow 2H_2WO_4$ . As reported in in vivo and in vitro studies of other types of bioresorbable <br/>electronic devices, these electrode materials harmlessly degrade and eventually disappear completely.(26, 27) Patterning of the electrodes (Mg, Zn, Fe, Mo, W) relies on a laser cutting process applied to metal foils. The MoO<sub>3</sub> cathodes result from casting of slurries of MoO3 on laser-cut Mo foils. Discharging bioresorbable <br/> batteries formed with these electrodes by immersing them in PBS reveals their behaviors. As shown in Fig. 2B and SI Appendix, **Fig. S6**, Mg-MoO<sub>3</sub>, Zn-MoO<sub>3</sub>, Mg-Fe, Mg-Mo, and Mg-W cells denerate an average voltage output of ~1.48 V, ~0.90 V, ~0.38 V, ~0.15 V, and ~0.13 V, respectively, and with an areal capacity in the range of 0 to 1.5 mAh cm<sup>-2</sup>, when discharged at a constant current density (CCD) of 0.4 mA cm<sup>-2</sup>. Due to the relatively high electropositivity of Zn, the devices formed with Zn-Fe, Zn-Mo, and Zn-W pairs show voltage outputs of ~0 V. These combinations are not considered further. Performing a shortcircuit discharge at a constant voltage (CVD) of 0 V represents the self-powered operation scenario that follows from exposure of the phototransistors to light (Fig. 2C). The Mg-MoO<sub>3</sub>, Zn- $MoO_3$ , Mg-Fe, Mg-Mo, and Mg-W cells offer an average short-circuit current of 93, 85, 31, 24, and 10 mA cm<sup>-2</sup>, respectively. For a specific metal anode (Mg or Zn), a high rate of discharge corresponds to a fast rate of gate corrosion and therefore a short triggering period for drug release. The Mg gate corrodes at an average rate of ~4.8, ~1.3, ~1.0, and ~0.4  $\mu$ m/min when paired with MoO<sub>3</sub>, Fe, Mo, and W cathodes, respectively. The Zn gate corrodes at an average rate of ~1.7  $\mu$ m/min in a Zn-MoO<sub>3</sub> cell setup (Fig. 2D). SI Appendix, Fig. S7 and Note 1 present calculations of the capacity required for complete corrosion of the Mg and Zn gates. Increasing the size of the cathode enhances the rate of gate corrosion (SI Appendix, Fig. S8 A and B), thereby justifying the use of the large cathode in the device designs presented here. Four tests of the short-circuit process using the same Fe cathode and fresh Mg anodes reveal negligible differences in the current densities. This result confirms the ability of the Fe cathode to support at least three consecutive drug-release events (SI Appendix, Fig. S8C).

Mg, as the most electroactive bioresorbable metal, degrades at a rate of as high as ~1.2 to 12  $\mu$ m/d in simulated body fluids and at physiological conditions (26). This timescale limits the lifetimes of devices that use Mg gates. Forming a coating of a polysiloxane-based spin-on-glass dielectric material (SOG, ~500 nm thickness) on the Mg foils enhances their stability in biofluids (44). This type of SOG is biocompatible and bioresorbable, as reported elsewhere (45). As determined by X-ray photoelectron spectroscopy, the chemical formula is  $SiO_{1.8}C_{0.6}$  (*SI Appendix*, Fig. S9 *A* and *B*). The CH<sub>3</sub>, Si-CH<sub>3</sub>, and Si-O-Si bonds (46–49) identified by Fourier transform infrared spectroscopy are consistent with a silicon oxycarbide structure (SI Appendix, Fig. S9C). Scanning electron microscopy (SEM) shows a representative uniform coating of SOG on a Mg foil (SI Appendix, Fig. S9D). SOG dissolves at a rate of ~1 nm/d in 37 °C PBS (SI Appendix, Fig. S10). Performing linear sweep voltammetry (LSV) measurements through a three-electrode setup in PBS with Mg, Mg-SOG, or Zn as the working electrode, a Pt wire as the counter electrode, and Ag/AgCl as the reference electrode, indicates that the natural corrosion rates of Mg, Mg-SOG, and Zn are 20, 0.03, and 0.18  $\mu$ m/d, respectively (Fig. 2*E*). The SOG coating greatly reduces the rate of the pure Mg metal, as expected. Coatings of SiO<sub>2</sub> (100 nm or 1 µm thick) formed by sputter deposition (Mg-SiO<sub>2</sub>) also reduce the rates, but by a comparatively lesser extent, possibly due to defects in the SiO<sub>2</sub> as indicated in SEM images (SI Appendix,

Fig. S11). Additional short-circuit discharge tests of cells with Mg-SOG or Mg-SiO<sub>2</sub> anodes and a Fe cathode confirm the ability of these anodes to function as gates. The case of Mg-SiO<sub>2</sub> shows similar discharge profiles to those with pure Mg, due to limited protection provided by the sputtered SiO<sub>2</sub>. The Mg-SOG system behaves differently, however, with a low initial short-circuit current of  $<2 \text{ mA cm}^{-2}$  and a gradual increase to 15 mA cm $^{-2}$ , equivalent to the case of pure Mg, within 30 s (SI Appendix, Fig. S12A). Electrochemical impedance spectroscopy (EIS) measurements conducted through a three-electrode setup in PBS with Mg-SOG as the working electrode, a Pt wire as the counter electrode, and Ag/AgCl as the reference electrode provide some insights. EIS measured before and after short-circuiting electrochemical cells with Mg-SOG anodes and Fe cathodes (SI Appendix, Fig. S12B) show that the resistance of the SOG coating, more specifically the resistance of the defects (50), decreases from  $1.2 \times 10^6$  to  $2.8 \times$ 10<sup>4</sup> ohm within 5 s after short-circuiting. Optical microscopy shows that this short-circuit discharge quickly destroys the SOG coatings as a result of pinholes that give rise to hydrogen bubbles generated from the corrosion reactions of the underlying Mg. These phenomena explain how the Mg-SOG gate corrosion can be triggered within a few seconds after activation in a short-circuit condition (SI Appendix, Fig. S12C). Corrosion rates of Mg-SOG recorded after its incubation in 37 °C PBS (SI Appendix, Fig. S13) provide data on the long-term protective ability of the SOG on Mg. Results show that the corrosion rate increases from 0.03  $\mu$ m/d at day 0, to 0.2 and 1.4 µm/d at day 1 and day 17, respectively, likely due to pinholes in the SOG coating. The protection provided by the SOG thus extends the lifetime of 50-µm-thick Mg gates from 7 to 23 d (Fig. 2F), as confirmed by recording the leakage of water with dye (Rhodamine B) loaded into reservoirs sealed with Mg and Mg-SOG gates at 37 °C in PBS, as measured by ultravioletvisible (UV-VIS) spectroscopy. Zn gates (10 µm) exhibit lifetimes of ~66 d, consistent with their natural rates of corrosion (0.18  $\mu$ m/d, Fig. 2*E*). Triggering a drug release device formed with a 20-µm-thick Mg-SOG gate paired with a Fe cathode for ~10 min opens (defined as >1% drug release) the gate. The cumulative release of drug reaches 90% in 30 min, 95% in 1 h, and 99% in 2 h (Fig. 2G).

Programmability of the Drug Delivery Device. As described previously, programmability relies on a combination of phototransistors/filters and light sources with narrow spectral emission aligned with the transmission properties of the filters, in a form of wavelength-division multiplexing. Descriptions of specific choices for demonstrations reported here appear in SI Appendix, Supplementary Note 2 and Fig. S14. The attenuation of light from LEDs at different implantation depths are calculated through a simplified Beer-Lambert law (51-53) (SI Appendix, Fig. S15). Light emitted from LED X selectively passes through optical filters X, illuminates phototransistor X, and triggers corrosion of Gate X (X = I, II, or III). In this sense, LED X, phototransistor X, Filter X, and Gate X form a matched group. The type of phototransistor used in this work has a dark current of 50 nA, corresponding to an "off" state resistance of 2 Mohms. This resistance decreases sharply upon increasing illumination, until reaching saturation at resistances of ~174, ~172, and ~91 ohms for illumination intensities of ~27, ~15, and ~64 mW cm<sup>-2</sup>, by light from LEDs I, II, and III, respectively. The rate of gate corrosion depends on both the resistance of the phototransistor and the resistance of the electrochemical cell. The ideal scenario involves illumination from LED X at an "on" state intensity sufficiently high such that the resistance of the phototransistor X is much lower than that of the cell (~1 kohms), but sufficiently

low to avoid crosstalk between other phototransistor/filter sets. For the system reported here, the intensities from light emitted by LEDs I, II, and III at the location of the phototransistor/filter sets are below 10, 10, and 2 mW cm<sup>-2</sup>, respectively, to ensure reliable programmability. As shown in SI Appendix, Fig. S16 A and *B*, the resistances of phototransistors underneath the matched filters are 62, 36, and 33 times smaller than that underneath the unmatched ones, for illumination by light from LEDs I, II, and III, respectively. For a three-reservoir drug device with Mg gates paired with a Fe cathode (Fig. 3A), the short-circuit current for the gate that connects to the phototransistor underneath the matched filter is > 25, > 11, and > 7 times that of the unmatched ones, for illumination by light from LEDs I, II, and III, respectively, which eliminates unwanted drug leakage from unmatched reservoirs, and ensures reliable programmability of release (SI Appendix, Fig. S16 C-E). As described before, a high short-circuit current corresponds to a fast gate-opening process. Finite element analysis (FEA, Fig. 3B) shows the current distributions of gates under 10, 10, and 2 mW cm<sup>-2</sup> light emitted from LEDs I, II, and III, respectively. Results indicate crevice corrosion processes, where the edges of the gates exhibit the highest corrosion rates, consistent with previous reports (15, 54). Fig. 3C presents a demonstration of programmable drug release, triggered by illuminating LEDs I, II, and III sequentially.

**In Vivo Drug Delivery and Biocompatibility.** Analgesic medications relieve pain from injuries due to trauma or medical procedures. A drug release device of the type introduced here can support programmable blocking of pain signals (Fig. 4 *A* and *B*), where

the device ultimately dissolves in the body to avoid the need for surgical extraction after a period of use (Fig. 4C). Fig. 4D illustrates implantation with the gates placed adjacent to the sciatic nerve of a rat model. The reservoirs in this case contain lidocaine for local pain block through preferential binding to the inactivated state of the voltage-gated sodium channels (55). Fig. 4E presents a photograph that shows the site for implantation. The operation of three LEDs triggers programmable release of lidocaine from one or multiple reservoirs. Compound muscle action potentials (CMAPs) and sensory nerve action potentials (SNAPs) measured before and after release quantify the efficacy of nerve block (Fig. 4 F and G). The release from a single reservoir leads to decreases in the amplitude of CMAP and SNAP by ~48% and ~55%, respectively. Releasing drugs from three reservoirs decrease the CMAP and SNAP amplitudes by ~87% and ~86%, respectively. The maximum block occurs within 15 min after release and nerve activity gradually recovers over the next ~1 h (SI Appendix, Fig. S17). To confirm the physiological effect of the lidocaine being released from the device, we measured the grip strength and Von Frey paw withdrawal threshold of both hindlimbs (Fig. 4 H and I and SI Appendix, Fig. S18) before device implantation (baseline), after waking up from lidocaine release to the right sciatic nerves of rats (drug), and after 24 h recovery (recovery). The release of lidocaine to the right sciatic nerve induces a decrease in grip strength and an increase in paw withdrawal threshold of the right hindlimb, which confirmed the physiological effect of the lidocaine being released from the device. The grip strength and paw withdrawal threshold of the left (contralateral) hindlimb were measured as well to verify that the rats woke up completely from



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general anesthesia. In this manner, the device can be utilized for programmed, on-demand pain relief during and after recovery from an injury or a surgical operation.

In vitro cytocompatibility studies of the device against human cells used both direct and indirect contact methods. The results indicate cytocompatibility of the device (*SI Appendix*, Fig. S19). Further evaluation of the biocompatibility of the gates and the devices involves subdermal implantation for 1-mo of a control group [PA (29), n = 4], a gate group (a Mg-SOG foil on PA, n = 4), and a device group (Mg-SOG and Fe foils on PA, n = 4), in the interscapular region of mice. The results in Fig. 5 A and B show negligible differences in the overall weight and changes in the weights of organ in mice throughout the gate, device, and control groups. Organs (brain, heart, lung, liver, spleen, and kidney) collected at 1- and 4-wk endpoints yields information on the biodistribution of key bioresorbable elements (Mg, Si, and Fe). Fig. 5C shows concentrations of Mg, Si, and Fe in organ tissues, measured by inductively coupled plasma optical emission spectrometry (ICP-OES). Device and gate groups show no abnormal accumulation of dissolved Mg, Si, and Fe in organ tissues compared to control group (SI Appendix, Tables S5 and S6 for analysis). SI Appendix, Supplementary Note

4 discussed the possible reasons (56–58) for elemental changes in spleen (for Si, Fe, SI Appendix, Fig. S20) and heart (for Si), from 1-wk to 4-wk endpoints. The analysis of complete blood counts and blood chemistry tests reveal no statistically significant differences among control, gate, and device groups (significance level P < 0.05, One-Way ANOVA), suggesting no device-related organ toxicity or disruption of electrolytes (chloride, Cl; sodium, Na; bicarbonate, TCO2; phosphorus, PHOS; calcium, Ca; potassium, K) and enzymes (aspartate transaminase, AST; alkaline phosphatase, ALP; both are indicators for liver function) (Fig. 5D). The blood urea nitrogen values increased from ~19.7 mg/dL at 1 wk to ~28.2 mg/dL at 4 wk, which falls within the normal range of mice (18 to 29 mg/ dL)(59) and might be attributed to the normal biological variability. Hematoxylin and eosin (H&E) images of stained tissue sections collected appear in Fig. 6 A and B and SI Appendix, Fig. S21, which indicates no damage to organ, skin, and muscle tissues and minimal to mild inflammatory cells related to implantation, consisting of loosely arranged histiocytes, neutrophils, and fibroblasts. At the end of 1 wk, small numbers of fibroblasts are present, indicating an early reparative process. At the end of 4 wk, the fibroblasts are more organized and



Fig. 5. In vivo biocompatibility and toxicity studies of Mg-SOG gates and full devices. (A) Change in the weight of might be determined by the studies of t

accompanied by deposits of loose fibrous tissue, consistent with a more organized inflammatory and reparative response to the implanted device. Dense fibrous capsule do not appear in any of the animals, suggesting a low severity of the inflammation and the absence of a robust immunologic response directed against the implanted device. This foreign body response might affect drug delivery process, which could be mitigated through pharmacological strategies [e.g., inflammasome inhibitor (60)]. Careful examination of H&E slides for fibrosis and inflammation in a five-grade scoring system (SI Appendix, Figs. S22 and S23) confirms the biocompatibility of the constituent materials used in this platform, consistent with previous reports (26, 27, 45). Additional biocompatibility studies involve triggering short-circuit of the device, a process that evolves hydrogen. Results (SI Appendix, Fig. S24) indicate that the device with hydrogen evolution during a short triggering period (~10 min) does not elicit a strong immune response and does not result in the presence of a dense fibrous capsule. Gas cavities are not present when removing devices at 1- and 4-wk endpoints, which could be attributed to hydrogen escape through dissolution into the adjacent tissues, diffusion into tissues and blood vessels, and excretion through skin (40, 43). Detailed calculations are in SI Appendix, Supplementary Note 3.

#### Discussion

This paper introduces a self-powered, light-controlled gate valve mechanism inspired by battery short-circuit phenomena and associated corrosion chemistry. A miniaturized drug delivery device built with bioresorbable materials in a multiple reservoir geometry serves as a demonstration vehicle for programmable operation. Here, phototransistors with integrated optical filters act as optically controlled switching elements to activate corresponding gates by illumination with light-emitting diodes. Animal model experiments illustrate the use of this system for on-demand pain relief and provide data on the aspects related to biocompatibility and bioresorbability. The limitations of this work include 1) the use of non-resorbable phototransistors and optical filters and 2) the practical constraints on the number of reservoirs that can be independently addressed using our optical strategy. On-going research focuses on designs for resorbable components to address the first limitation (61). For the second, given angular uncertainties and optical scattering by biological tissues, optical filters with transmission bandwidths less than ~20 nm (64) will be ineffective. Considering a realistic range of wavelengths based on tissue absorption and subdermal location of photodetectors of the system, i.e., 400 nm to 1,000 nm, this transmission bandwidth limitation



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suggests an ability to independently control no more than 30 reservoirs. Spatial selectivity in illumination might provide a route to further increase the number of reservoirs. Adapted versions of the platforms reported here also have potential for chronic treatments and cocktail therapies, as additional areas for future work.

## **Materials and Methods**

**Fabrication of Transient Batteries.** An ultraviolet laser prototyping system (LPKF U4) defined the shapes of the anodes and cathodes from foils of Mg (50 or 100  $\mu$ m, Alibaba), Zn (10 or 20  $\mu$ m, Alibaba), Mo (5 or 50  $\mu$ m, Alibaba), and W (50  $\mu$ m, Alibaba) and from foils of Fe (3 or 25  $\mu$ m, Alfa Aesar). Mg-Fe, Mg-Mo, and Mg-W batteries incorporated anodes of Mg, cathodes of Fe, Mo, or W, and electrolytes of phosphate buffered solution (PBS, Sigma-Aldrich). Mg-MoO<sub>3</sub> and Zn-MoO<sub>3</sub> batteries combined anodes of Mg or Zn with MoO<sub>3</sub> cathodes, respectively, in PBS electrolytes. Mixing 0.45 g MoO<sub>3</sub> (Sigma Aldrich), 0.05 g carbon black, and 800  $\mu$ L of 300 mg/mL poly (D,L-lactide-co-glycolide) (PLGA, lactide/glycolide 65:35, M<sub>w</sub> 40 to 75k, Sigma-Aldrich) in ethyl acetate within a planetary mixer (Thinky ARE-30) yielded a slurry for the MoO<sub>3</sub> cathode. Mo gauze (40 mesh, Alfa Aesar) and Mo foil (5 or 50  $\mu$ m) served as current collectors for the cathode materials. Applying the slurries on these current collectors and baking at 65 °C for 30 min completed the process.

**Electrochemical Measurements.** Methods for discharging the transient batteries used two different conditions 1) constant current discharge (CCD) at 0.4 mA cm<sub>Anode</sub><sup>-2</sup> using the chronopotentiometry technique, and 2) constant voltage discharge (CVD) at 0 V using the chronoamperometry technique. Linear sweep voltammetry (LSV) and electrochemical impedance spectroscopy (EIS) measurements involved a three-electrode setup in PBS with Mg or Mg-SOG as the working electrode, Pt wire as the counter electrode, and Ag/AgCl as the reference electrode. For LSV, the scanning potential ranged from -250 mV to 250 mV at a rate of 1 mV/s. EIS under galvanostatic mode used a frequency range of 10 kHz to 0.1 Hz, with a current amplitude of 0.1  $\mu$ A. All electrochemical measurements relied on a PalmSens4 system.

Forming Coatings of Spin-on-Glass (SOG) and SiO<sub>2</sub> on Mg Foils. Spin casting (1,000 rpm for 10 s) a polysiloxane-based spin-on dielectric material (IC1-200, Futurrex) on Mg foils, followed by baking at 250 °C for 2 h, yielded coating of SOG with thicknesses of ~500 nm (Mg-SOG). Repeating this process formed coatings on both sides of the foils. Forming films of SiO<sub>2</sub> with thicknesses of 100 nm and 1  $\mu$ m by sputter deposition onto Mg foils yielded SiO<sub>2</sub>-coated Mg foils (Mg-SiO<sub>2</sub>).

Material Characterization. Morphological and compositional analyses used scanning electron microscopy (SEM, Hitachi SU 8030). A Thermo Scientific ESCALAB QXi XPS Microprobe-enabled X-ray photoelectron spectroscopy (XPS)

measurements. Fourier transform infrared spectroscopy (FTIR) used a Nexus 870 spectrometer. A Filmetrics F20 instrument yielded the thicknesses of the layers of SOG. A Keyence microscope captured all optical micrographs.

Fabrication of Drug Delivery Devices. Polybutanedithiol 1,3,5-triallyl-1,3,5-triazine-2,4,6(1H,3H,5H)-trione pentanoic anhydride (PA)(29) served as the material for the drug reservoirs (total size of 19 mm × 13 mm × 1.25 mm for three reservoirs), synthesized through thiol-ene click reactions of three compounds: 4-pentenoic anhydride (4PA, Sigma-Aldrich), 1,3,5-triallyl-1,3,5-triazine-2,4,6(1H,3H,5H)-trione (TTT, Sigma-Aldrich), and 1,4-butanedithiol (Sigma-Aldrich). 4-PA, TTT, and 1,4-butanedithiol mixed with a molar ratio of 1:1:2.5 (for accelerated aging test) or 1:4:7 (for all other experiments) in Ecoflex 35 molds, followed by exposure to ultraviolet (UV, wavelength of 365 nm) light with 2,2-dimethoxy-2-phenylacetophenone (Sigma-Aldrich) as the photoinitiator (total mass of 0.5%). Disks of 5 mm × 5 mm Mg (20 to 50 µm, wet-etched), Mg-SOG (20 to 50 µm), and Zn (20 µm) defined by a laser cutting process served as the metal gates. Laser-cut 5 mm × 17 mm Mo and Fe foils served as cathodes. A flat film of PA (19 mm × 13 mm × 0.25 mm) sealed the reservoirs. Extended metal electrodes connected to a commercial phototransistor (Digi-Key, NTE3120A). Optical filters (Edmund Optics) placed atop the phototransistors provided a wavelength-selective response. Biodegradable conductive paste (Candelilla wax/W powder) (30) electrically connected the individual components. Three sets of optical filters and light-emitting diodes (Digi-Key) enabled programmable release of drugs from the reservoirs. A digital multimeter measured the resistance of a phototransistor placed underneath each optical filter and illuminated by each LED.

**Finite Element Analysis.** 3D FEA used the commercial software Ansys Maxwell to define the distributions of current density associated with the drug release devices. The use of an adaptive mesh (tetrahedron elements) ensured computational accuracy. The relative permittivity ( $\varepsilon_r$ ) and bulk conductivity ( $\sigma$ ) were 80 and 1.4 S·m<sup>-1</sup> for PBS, and 3 and 6.7\*10<sup>-16</sup> S m<sup>-1</sup> for PA. The electromagnetic parameters of other materials relied on the material library in the Ansys Maxwell software package.

**Measurements of Cumulative Release of Drug.** Tests involved a single drug reservoir with gates of Mg, Mg-SOG, or Zn and filled with 0.005 M Rhodamine B solution, triggered to release this solution into surrounding PBS or incubated at 37 °C in PBS to investigate the release process through different gates. At regular time intervals, 3 mL of surrounding PBS was drawn from the vial for UV-VIS spectroscopic analysis (HP8452, Agilent Technologies), subsequently returned to the vial after each measurement. The concentration of Rhodamine B followed from measurements of the absorption peak at 554 nm.

In Vivo Demonstration of Programmable Lidocaine Release for Sciatic Nerve Block. Drug delivery devices with three drug reservoirs filled with lidocaine (Xylocaine-HCL 2%, Astra Zeneca Pharma) were surgically implanted onto the right sciatic nerves of female Sprague-Dawley rats (200 to 250 g, Charles River Laboratories). All animal procedures were performed in accordance with the ethical standards and protocols of Northwestern University. During implantation, rats were anesthetized using isoflurane gas (3% for induction and 1 to 2% for maintenance). After the implantation, the muscle was closed with 6-0 Vicryl sutures and the skin incision was closed with Autoclip wound clips (12020-09, Fine Science Tools). Three LEDs placed over the implantation site served as the basis for programmable triggering of the release of lidocaine from anyone or any arbitrary combination of reservoirs. CMAP and SNAP were measured with 1" length, 0.25-mm diameter concentric needle electrodes (UTC-25F, The Electrode Store) on a VikingQuest EMG system using Synergy master software (Natus). (Data are means  $\pm$  SD; n = 3.) To confirm the physiological effect of the lidocaine being released, devices (filled with  $\sim$ 230  $\mu$ L lidocaine) were implanted adjacent to the right sciatic nerves of rats followed by triggering the drug release process under anesthesia. After 30 min, these rats completely woke up. Hindlimb strength measurements were made with a digital grip strength meter with mesh assembly (Columbus Instruments, Columbus, OH) with the average of 3 to 5 trials recorded. Nociception function was assessed by recording the paw's withdrawal threshold from mechanical stimulation using an electronic Von Frey system (Harvard Apparatus, Holliston, Massachusetts) with the average of 3 to 5 trials recorded.

In Vitro Cytocompatibility. In vitro cytocompatibility studies of the device against human cells (Normal human dermal fibroblasts, NHDF, CC-2511, Lonza) were performed through both direct and indirect contact methods. A mixture of fibroblast basal medium (ATCC, CAT# PCS-201-030), 10 vol.% fetal bovine serum (Thermo Fisher Scientific) and 1 vol.% penicillin-streptomycin (Thermo Fisher Scientific) served as the cell culture medium. In direct contact experiments, samples were incubated in the cell-seeded wells for 24 h. In indirect contact experiments, incubating samples in cell medium for 24 h resulted sample extracts (ISO 10993-12), which were centrifuged at 14,000 rpm for 10 min. Supernatants were added into the cell-seeded wells for 24 h. Cells stained with a live/dead viability/ cytotoxicity kit (L3224; Thermo Fisher Scientific) according to the manufacturer's protocols allowed the quantitative evaluations of the cytocompatibility of samples, including control (PA) and device groups.

In Vivo Biocompatibility Tests. All procedures were performed in accordance with the ethical standards and protocols of Northwestern University. Each male CD-1 mouse (33 to 38 g) was placed under isoflurane anesthesia (~3%), then given pre-operative analgesia (SQ: 20 mg/kg Meloxicam, line block of 2 mg/ kg - 0.1% Bupivacaine). Under sterile conditions, a ~0.7-cm incision was made on the back. Two types of devices (PA-Mg-SOG, n = 4, and PA-Mg-SOG-Fe, n = 4) and a control group (PA, n = 4) were implanted subcutaneously in the interscapular region. In group PA-Mg-SOG-Fe, Mg, and Fe were electrically connected by biodegradable conductive W paste. Mice were weighed and checked at regular time intervals. Blood, skin, muscle adjacent to the device, and organs (brain,

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heart, lung, liver, spleen, and kidney) were collected at 1- and 4-wk endpoints. Skin, muscle, and organ tissues were fixed in 10% neutral-buffered formalin, embedded in paraffin, and stained with hematoxylin and eosin (H&E) for histological analysis. Concentrations of Mg, Si, and Fe in the organs were measured by inductively coupled plasma optical emission spectrometry (ICP-OES).

Histological Scoring. The histologic evaluation of the tissue was performed and a five-grade scoring system was used to record the histopathology severity. Each H&E slide was carefully reviewed for the presence of fibrosis, mesenchymal cells, hematopoietic cells, basophilia and inflammation. Severity was scored as 0 =none, 1 =minimal, 2 =mild, 3 =moderate, 4 =marked, and 5 =severe.

Data, Materials, and Software Availability. All study data are included in the article and/or SI Appendix.

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